

Characterization of defects in Ni alloys based on gamma-ray-induced positron annihilation lifetime spectroscopy at UVSOR

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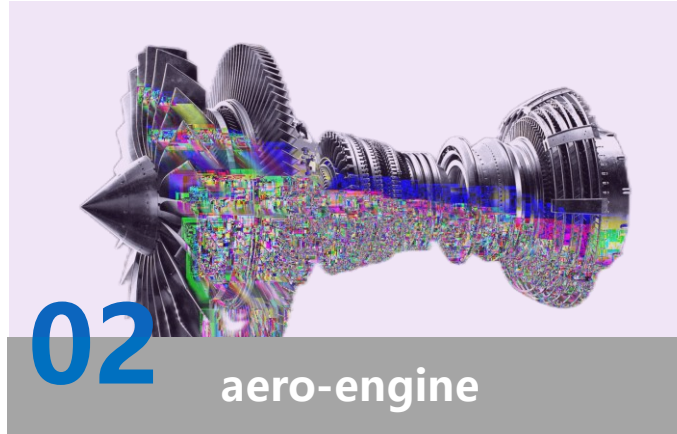
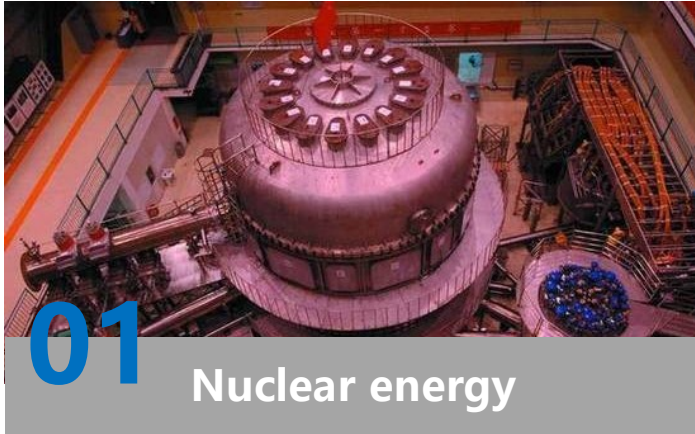
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OUTLINE

- Research motivation
- Experimental method
- Results
- Conclusions



MOTIVATION



Nickel based superalloys are widely used in various industrial applications, particularly in the hot sections, like nuclear energy installations, aero-engines, gas turbines, and other critical equipment. However, during long-term service under cyclic mechanical and thermal load, the stability of the superalloys may deteriorate as the formation of microscopic defects.



MOTIVATION

Å

Multi-scale defects

mm

High temperature, irradiation, stress

Point defects

Diffusion

Microstructural defects

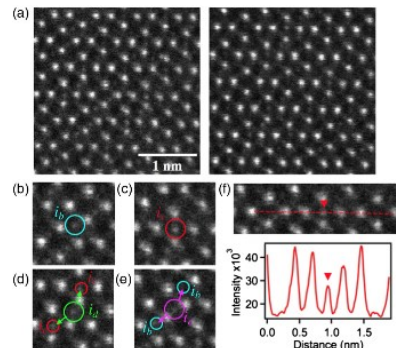
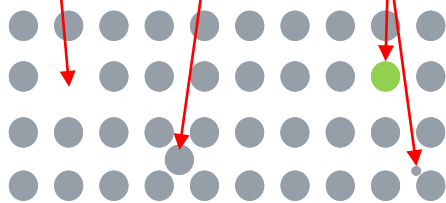
Diffusion

Mesostructural defects

Further development

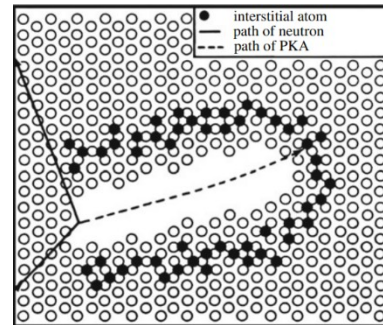
Macrostructural defects

vacancy interstitial impurity

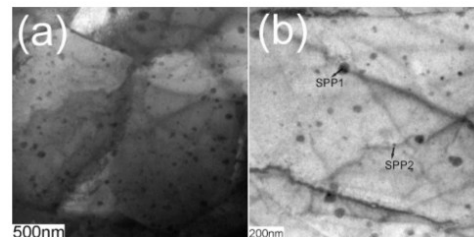


Direct detection of interstitial defects in Sn-doped β -Ga₂O₃

vacancy clusters

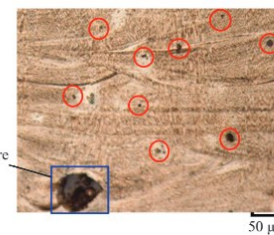


nano-precipitates



bright field (BF) image typical of the microstructure of the ZIRLO alloy

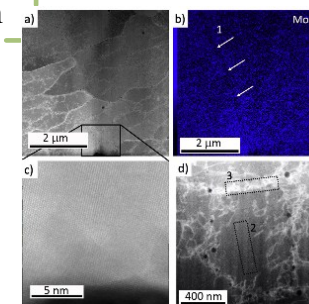
pore



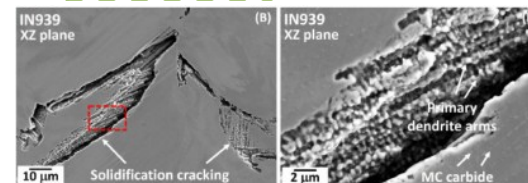
Small sized pores of 10-μm diameter and large-sized pores of more than 50-μm diameter.

dislocation

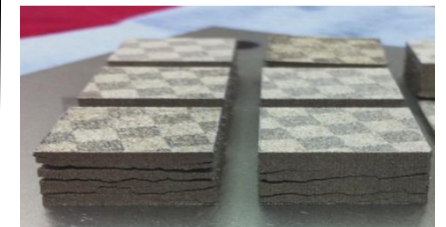
Direct detection of dislocation structure



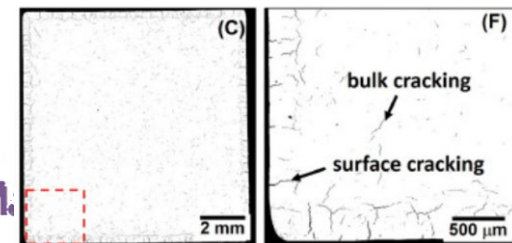
Microcracks



delamination

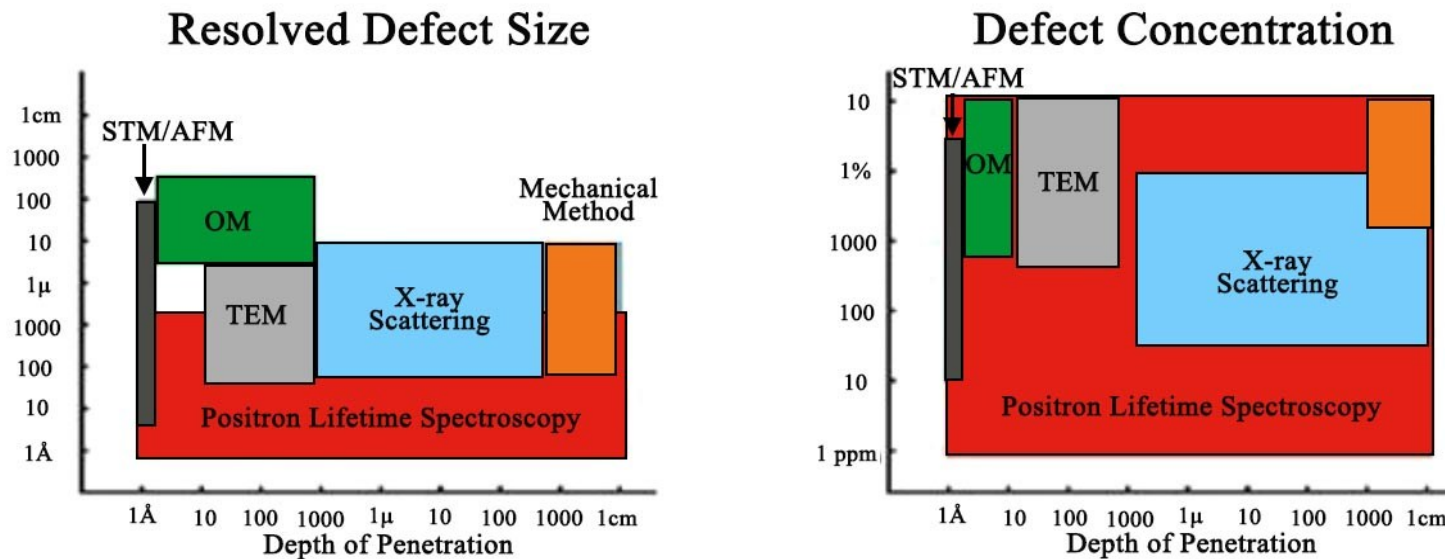


Surface cracks



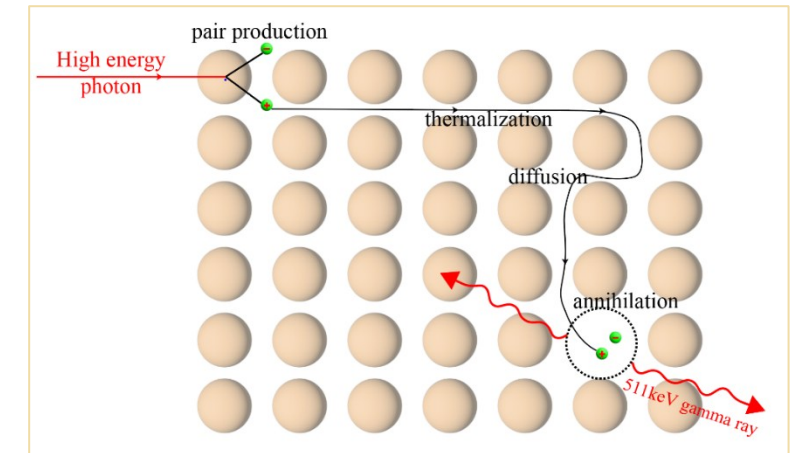
MOTIVATION

Characterization of defects with nanoscale microstructure is essential for accurately assessing the remaining service life.



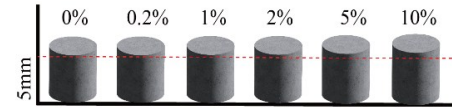
Comparison of the sensitivity of various techniques in defect measurements with respect to depth, size and concentration.

TEM: transmission electron microscopy,
STM: scanning tunneling microscopy,
AFM: atomic force microscopy,
OM: optical microscopy,



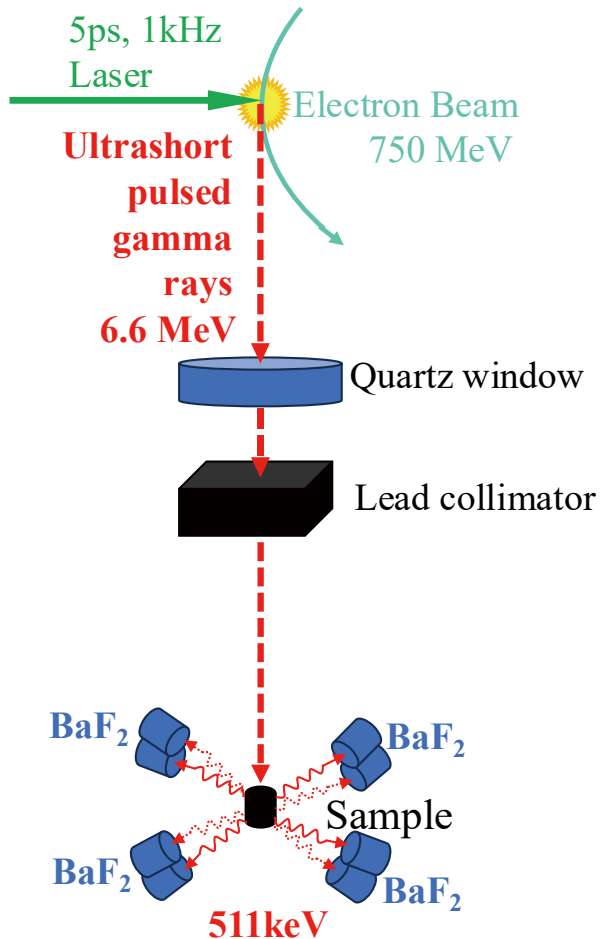
Experimental method

different deformation



Sample

Different deformation to simulate progressive different stages of defect development



convolution integral:
$$f(t) = \sum_{j=1}^{k_0} (a_j * R)(t) + B$$

where

$$a_j(t) = \begin{cases} A_j \exp(-(t - T_0)/\tau_j), & t > T_0 \\ 0, & t < T_0 \end{cases}$$

Positron lifetime component

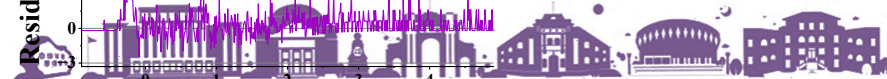
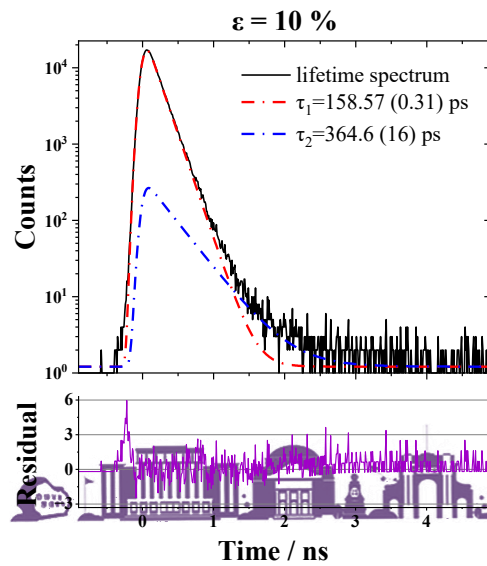
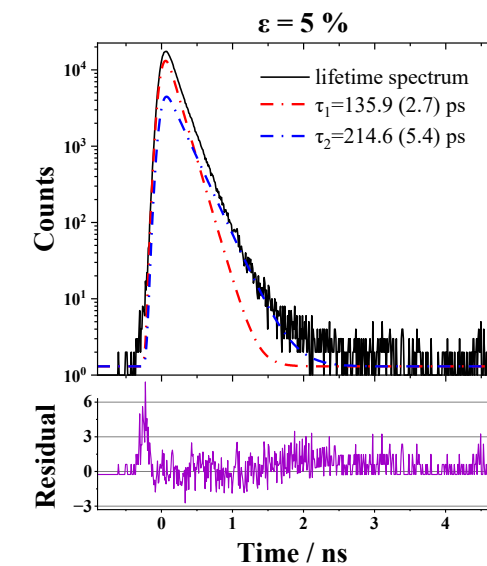
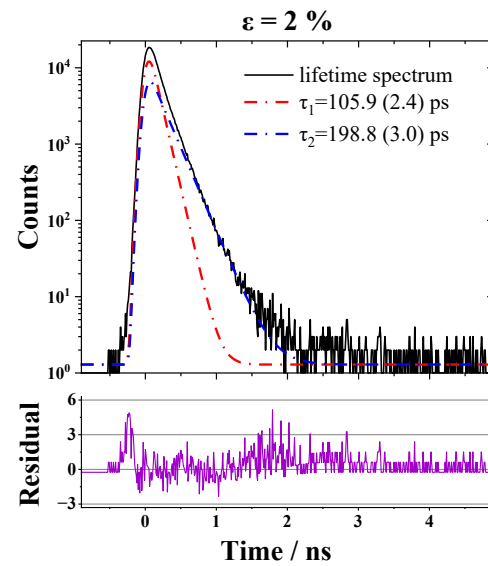
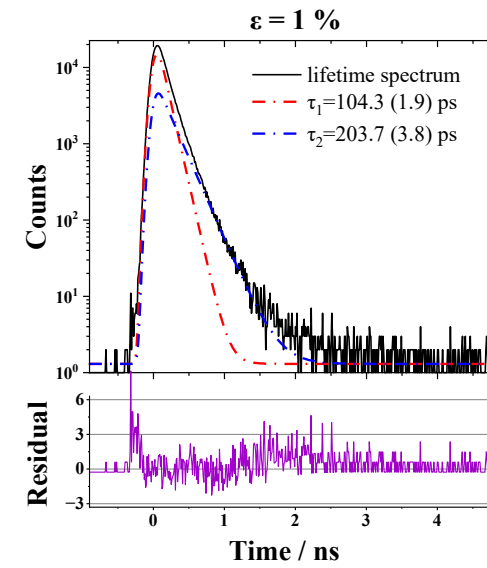
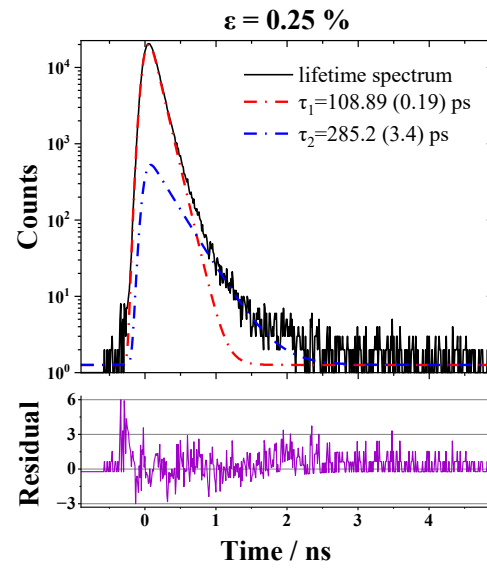
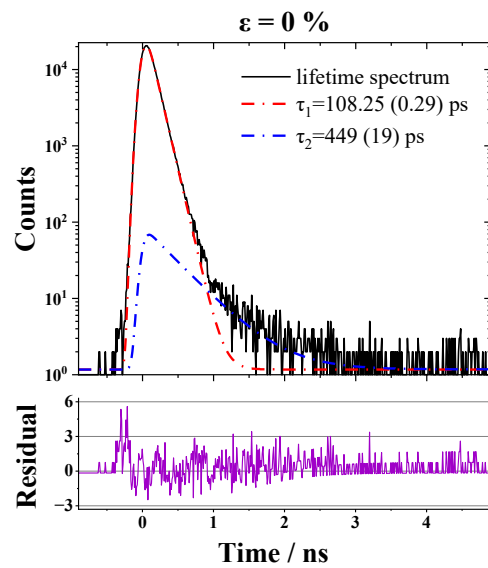
$$R(t) = \sum_{p=1}^{k_g} w_p G_p(t) = \sum_{p=1}^{k_g} w_p \frac{1}{\sqrt{2\pi} s_p} \exp\left(-\frac{(t - \Delta_p)^2}{2s_p^2}\right)$$

time-resolution function

$$\sum_{p=1}^{k_g} w_p = 1 \quad \int_{-\infty}^{\infty} R(t) dt = 1$$



Results

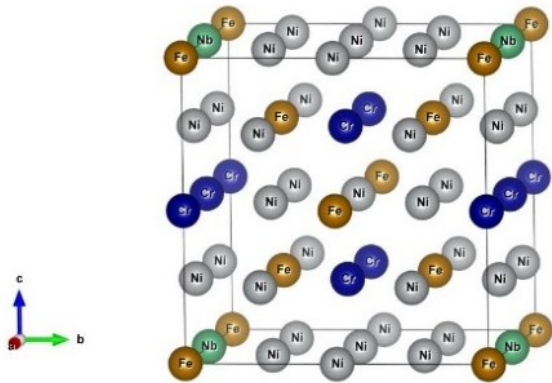


Results

$$\frac{1}{\tau} = \lambda = \pi r_e^2 c \int n(r) n_+(r) \gamma(r) dr$$



Calculation of positron annihilation lifetime

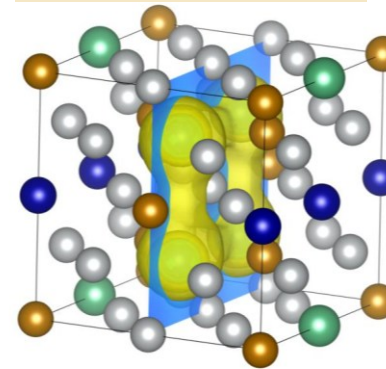


Ni₁₉Fe₆Cr₆Nb

a=b=c=7.1Å

Α=β=γ=90°

Perfect lattice



Vacancy-Ni

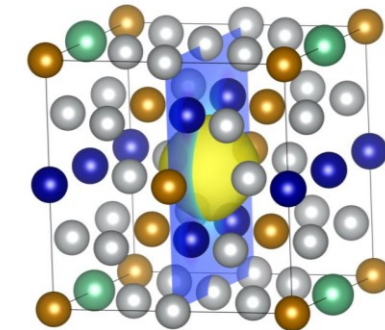
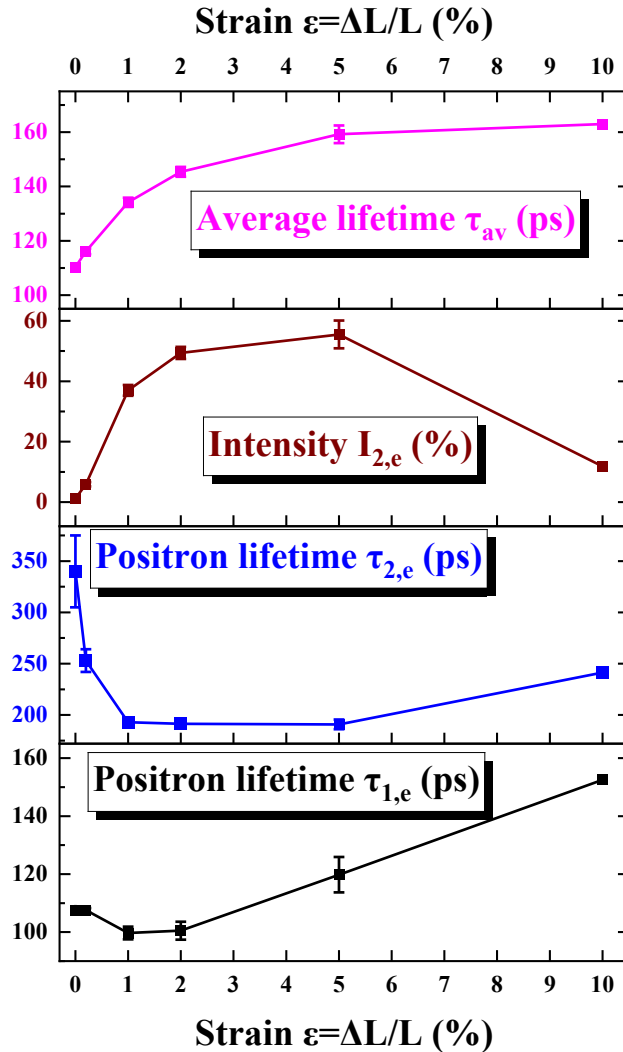


Table 1 Calculated positron state lifetimes (ps) for perfect lattice and defects in Ni₁₉Fe₆Cr₆Nb

Prefect lattice	Vacancy: Fe	Vacancy: Cr	Vacancy: Ni	Vacancy: Fe, Cr	Vacancy: Ni, Cr	Vacancy: Ni, Cr, Cr	Vacancy: Ni, Ni, Ni, Cr, Cr	Edge dislocation
105.7	160.5	184.1	164.0	178.4	181.1	190.3	224.5	145.5
Prefect	Vacancy							dislocation



Results

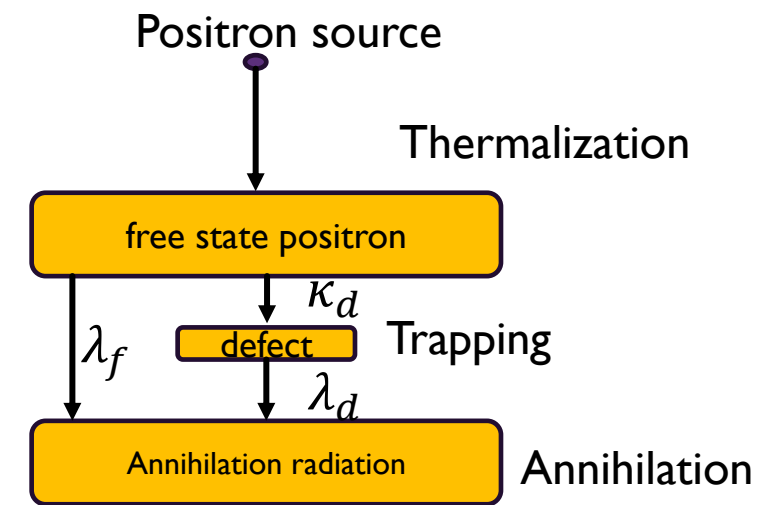


Trapping model

Assumption only one type defect in the sample:

$$\frac{dn_f(t)}{dt} = -\lambda_f n_f - \kappa_d n_f \quad (1)$$

$$\frac{dn_d(t)}{dt} = -\lambda_d n_d + \kappa_d n_f \quad (2)$$



n_f : the probability density of finding a delocalized (free) positron at the time t

n_d : the probability density of finding a localized positron at a defect at the time t

κ_d : the trapping rates to the defects in the system

λ_f : the free-positron annihilation rate

λ_d : the localized-positron annihilation rate.

The experimental spectrum can be described as:

$$S(t) = \lambda_f n_f + \lambda_d n_d = \left(\lambda_f - \frac{\lambda_d \kappa_d}{\lambda_f + \kappa_d - \lambda_d} \right) e^{-(\lambda_f + \kappa_d)t} + \frac{\lambda_d \kappa_d}{\lambda_f + \kappa_d - \lambda_d} e^{-\lambda_d t}$$

$$= \frac{I_1}{\tau_1} e^{-\frac{t}{\tau_1}} + \frac{I_2}{\tau_2} e^{-\frac{t}{\tau_2}}$$

$$\tau_1 = \frac{1}{\lambda_f + \kappa_d} \quad I_1 = \frac{1}{\lambda_f + \kappa_d} \left(\lambda_f - \frac{\lambda_d \kappa_d}{\lambda_f + \kappa_d - \lambda_d} \right) = \frac{\lambda_f - \lambda_d}{\lambda_f + \kappa_d - \lambda_d}$$

$$\tau_2 = \frac{1}{\lambda_d} \quad I_2 = \frac{\kappa_d}{\lambda_f + \kappa_d - \lambda_d}$$

CONCLUSIONS

- Two types of defects, **dislocations and vacancies**, can be identified with the aid of density functional theory.
- The observed increase in positron average lifetime with incremental tensile deformation can be used to characterized the defects in engineering materials for **industrial use**.
- Our work also provides valuable information on the **defect formation** of deformed samples contributing to a better understanding of fatigue damage.





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THANKS FOR YOUR ATTENTION!